# Magnetic and Magnetocaloric Properties of

## $_{0.5} Pr_{0.67} Ba_{0.22} Sr_{0.11} Mn_{0.95} Fe_{0.05} O_3/_{0.5} Pr_{0.67} Ba_{0.22} Sr_{0.11} Mn_{0.9} Fe_{0.1} O_3$

## Composite

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In the present work, the magnetic and magnetocaloric properties of  $_{0.5}Pr_{0.67}Ba_{0.22}Sr_{0.11}Mn_{0.95}Fe_{0.05}O_3$ / $_{0.5}Pr_{0.67}Ba_{0.22}Sr_{0.11}Mn_{0.9}Fe_{0.1}O_3$  (0.5PBSMF<sub>0.05</sub>/0.5PBSMF<sub>0.1</sub>) composite have been investigated. Our composite 0.5PBSMF<sub>0.05</sub>/0.5PBSMF<sub>0.1</sub> has been synthesized using the solid-state reaction. X-ray diffraction (XRD) at room temperature has been used to carry out the structural properties. In order to investigate the magnetic and magnetocaloric properties of the samples, magnetization measurements dependence on temperature and magnetic field have been performed by using physical property measurement system. XRD results indicate that all our simples crystallized in the orthorhombic structure with Pnma space group. Magnetization measurements versus temperature for the composite reveal two magnetic entropy change peak. Based on Maxwell relation, the maximum magnetic entropy change ( $|\Delta S_M^{max}|$ ) of the composite has been calculated as  $1.18 \times 10^{-3}$  Jkg<sup>-1</sup>K<sup>-1</sup>, which corresponds to relative cooling power (RCP) about 17.8 Jkg<sup>-1</sup> under a magnetic field change of 5T.

## 1. Introduction

Perovskite manganites with the general formula  $A_{1-x}B_xMnO_3$  (A = Pr, La, Nd,... and B = Ba, Sr, Ca,...) have been of large recent interest due to their electrical and magnetic properties [1-4]. This type of materials can be used in different technological applications such as colossal magneto-resistance, computer memory systems, storage devices and magnetic refrigerants [5-8]. The magnetic ground state in these materials is sensitive to variations of the Mn-O bond length and Mn-O-Mn bond angle and Mn<sup>3+</sup>/ Mn<sup>4+</sup> ions [9]. The variation of ions in A and B-sites has a sensitive impact on physical properties of perovskite manganites. Also, the Mn site substitution by different elements affects the Mn-O-Mn networks and the combination of the double-exchange interaction (DE) coupled Mn<sup>3+</sup>/ Mn<sup>4+</sup> ions [10, 11].

The reported magnetocaloric effect (MCE) in the manganites and the ability to control their Curie temperature by chemical substitutions opened doors for the production of new candidates of nonpolluting magnetic refrigerants in a wide temperature range. Also, MCE is becoming a promising technology to replace the conventional gas compression-expansion technique [12, 13]. The MCE can be defined as the adiabatic temperature change  $(\Delta T_{ad})$  or the isothermal magnetic entropy change  $(\Delta S_M)$  of a system due to the applied magnetic field [14]. In addition, the manganites have attracted the widest interest in aspects of theoretical and experimental researches due to their large magnetic entropy change. More than that, a large magnetic entropy change can be tuned in a large temperature range, intervening on the chemical composition of these materials, which is advantageous for magnetic refrigeration [15, 16]. Moreover, the manganites are very convenient for the synthesis routes, and their Curie temperature can be justified under various doping conditions. Therefore, the new trends have been focusing on studying the MCE of manganite composites [17, 18].

In this work, we elaborated by the solid state method at high temperature, the composite  $Pr_{0.67}Ba_{0.22}Sr_{0.11}Mn_{0.95}Fe_{0.05}O_3/Pr_{0.67}Ba_{0.22}Sr_{0.11}Mn_{0.9}$  Fe\_{0.1}O<sub>3</sub> composite studied their magnetic and magnetocaloric properties.

### 2. Results and Discussion

The powder X-ray diffraction (XRD) patterns of our samples have been recorded at room temperature. Figures 1a-b) show the refined XRD

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**Figure 1.** Observed and calculated X-ray diffraction data and Rietveld refinement for **a**) Pr0.67Ba0.22Sr0.11Mn0.95Fe0.05O3 and **b**) Pr0.67Ba0.22Sr0.11Mn0.9Fe0.1O3. Vertical bars are the Bragg reflections for the space group Pnma. The difference pattern between the observed data and fits is shown at the bottom.

Figure 2 shows the XRD patterns of  $0.5Pr_{0.67}Ba_{0.22}Sr_{0.11}Mn_{0.95}Fe_{0.05}O_3/0.5Pr_{0.67}Ba_{0.22}Sr_{0.1}$  ${}^{1}Mn_{0.9}Fe_{0.1}O_3$  composite. From the Fig. 2, we can notice the evolution of the most intense peak, which reflects a broadening signature of composite formation. These results are similar to those reported by *E. Sellami et al.* [22].



Figure 2. Refined X-ray powder diffraction for  $0.5Pr_{0.67}Ba_{0.22}Sr_{0.11}Mn_{0.95}Fe_{0.05}O_3/0.5Pr_{0.67}Ba_{0.22}Sr_{0.1} \ _1Mn_{0.9}Fe_{0.1}O_3$  composite.

The magnetic properties of the PBSMF<sub>0.05</sub> and PBSFM<sub>0.1</sub> manganites and their composite were characterized by the magnetization measurement vs. temperature in the range 5-350 K under a magnetic applied field of 0.05T. Fig. 3a shows the temperature dependence of the magnetization M (T) for both parent compounds. We clearly observed a sharp transition from paramagnetic to ferromagnetic state with decreasing temperature. The Curie temperature Tc was determined from dM/dT plots shown in the inset of Fig. 3a. It is found to be 135K for PBSMF0.05 and 70 K for PBSFM0.1. The M(T) curves of the 0.5PBSMF<sub>0.05</sub>/0.5PBSFM<sub>0.1</sub> composite shown in Fig. 3b have two magnetic phase transitions which are in accordance with the PBSMF<sub>0.05</sub> and PBSFM<sub>0.1</sub> phase transition. At low temperatures when compared to the magnetization values of the PBSMF<sub>0.05</sub> and PBSFM<sub>0.1</sub> manganites, the magnetization value of the 0.5PBSMF<sub>0.05</sub>/0.5PBSFM<sub>0.1</sub> composite is smaller than that of individual phases. This decrease caused by interactions between different magnetic phases with different  $T_c$  [23].

The variation of the magnetization as a function of magnetic applied field up to 5T at several temperatures.was also measured for 0.5 PBSMF0.05 /0.5 PBSFM0.1 composites (Fig. 4). At low temperatures, the magnetization M increases sharply at low magnetic fields ( $\mu$ 0H < 0.5 T) and then saturates above 1 T. This result confirms the existence of ferromagnetic state at low temperatures for our composite.

In order to study the nature of magnetic phase transition, Banerjee criterion allows the determination of the order of the magnetic transition. It is based on the observation of the slope of the isotherm plots  $M^2$  vs.  $\mu_0 H/M$ , a positive or a negative slope indicates a second order or first order transition, respectively [24]. The Arrott plots



 $M^2$  vs.  $\mu_0 H/M$  for the studied of the composite is displayed in Fig. 5. The Arrott curves have a positive slope witch indicate that our composite undergoes a second order FM–PM phase transition.



**Figure 3.** M(T) curves for **a)** PBSMF<sub>0.05</sub>, PBSFM<sub>0.1</sub> manganites and **b)** their composite at  $\mu$ OH = 0.05 T magnetic field. The inset is the dM/dT curves.



Figure 4. Isothermal magnetization for  $0.5Pr_{0.67}Ba_{0.22}Sr_{0.11}Mn_{0.95}Fe_{0.05}O_3/0.5Pr_{0.67}Ba_{0.22}Sr_{0.1}$  $_1Mn_{0.9}Fe_{0.1}O_3$  composite measured at several temperatures.



Figure 5. H/M vs.  $M^2$  curves of isotherms for  $0.5Pr_{0.67}Ba_{0.22}Sr_{0.11}Mn_{0.95}Fe_{0.05}O_3/0.5Pr_{0.67}Ba_{0.22}Sr_{0.1}$  $_1Mn_{0.9}Fe_{0.1}O_3$  composite.

The magnetic entropy change ( $\Delta S_M$ ) of the 0.5 PBSMF<sub>0.05</sub> /0.5 PBSFM<sub>0.1</sub> composite, can be calculated from the isothermal magnetization measurements by applying to the formula given below [25]

$$|\Delta S_M| = \sum_i \frac{M_i - M_{i+1}}{T_{i+1} - T_i} \Delta H \tag{1}$$

Where  $M_i$  and  $M_{i+1}$  are the values of magnetization measured at  $T_i$  and  $T_{i+1}$ , respectively.

Figure 6 shows  $|\Delta S_M|$  as a function of temperature under different magnetic applied field variations for the 0.5 PBSMF<sub>0.05</sub> /0.5 PBSFM<sub>0.1</sub> composite. As expected,  $|\Delta S_M|$  exhibits two maximums around the transition temperatures, and the maximum  $\Delta S_M$  value, increases with the amplitude of the magnetic applied field change. Under a magnetic applied field change of 5 T, it amounts 1.18 10<sup>-3</sup> J kg<sup>-1</sup> K<sup>-1</sup> for composite.



**Figure 6.** Temperature dependence of the magnetic entropy change  $|\Delta S_M|$  for the composite at different applied magnetic field.

The relative cooling power (RCP) for our composite, which expresses the cooling power of the magnetocaloric material, is determined by the following expression [26]

$$RCP(S) = -\Delta S_M^{max} \times \delta T_{FWHM}$$
(2)

 $\delta T_{FWHM}$  is the full-width at half maximum of the curve of  $\Delta S_M$ . Under a magnetic applied field of 5T, the obtained values of RCP are found to be 175, 122 and 17.8 J/Kg for the PBSMF0.05, PBSFM0.1 manganites and 0.5 PBSMF<sub>0.05</sub> /0.5 PBSFM<sub>0.1</sub> composite, respectively. The values display that RCP value of the composite is decreased by 10% when compared to those of PBSMF0.05, PBSFM0.1 manganites. The decrease of The RCP can be explained by the increase in the number ratio of Fe<sup>3+</sup>-O-Fe<sup>3+</sup> in the composite, which induces a weakening of the ferromagnetism. In addition, this evolution can be explained by the fact that the presence of iron weakens the double exchange interactions to the profit of the super exchange ones the complex Fe<sup>3+</sup>-O-Mn<sup>3+</sup>[27-31]. We in summarized in Table 1 the Curie temperature T<sub>C</sub>,  $-\Delta S_M^{max}$  and RCP values, in order to compare the results with those reported in literature [32-35].

**Table 1.** Comparison of magnetocaloric propertiesof our samples with the literature.

Materials	µ₀H	RCP	Ref.
	<b>(</b> T <b>)</b>	(Jkg <sup>-1</sup> )	
PBSMF <sub>0.05</sub>	5	175	This
			work
PBSFM <sub>0.1</sub>	5	122	This
			work
0.5 PBSMF <sub>0.05</sub> / 0.5	5	17.8	This
PBSFM <sub>0.1</sub>			work
La0.67Ca0.33MnO3/Fe	1	62.42	[32]
304			
(La0.7Ca0.3MnO3)0.9/	3	160	[33]
(Co <sub>3</sub> O <sub>4</sub> ) <sub>0.1</sub>			
0.9La <sub>0.6</sub> Ba <sub>0.2</sub> Sr <sub>0.2</sub> Mn	1	100.86	[34]
O <sub>3</sub> /0.1CO <sub>2</sub> O <sub>3</sub>			
Gd	5	410	[35]

## 3. Conclusion

To conclude our scientific studies, the 0.5 PBSMF<sub>0.05</sub> /0.5 PBSFM<sub>0.1</sub> composite has been obtained by the mixture of the PBSMF0.05 and PBSFM<sub>0.1</sub> samples in the mass ratio of 0.5:0.5 using the solid-state reaction. The magnetic and magnetocaloric properties have been studied by magnetization measurements. Magnetic measurements show that all samples exhibit a PM-FM transition with decreasing temperature. They exhibit а second-order ferromagnetic to paramagnetic phase transition. The magnetic phase transition of the materials has been determined as second order by using Banerjee criterion. The  $-\Delta S_M^{max}$  and RCP value for the 0.5 PBSMF<sub>0.05</sub> /0.5 PBSFM<sub>0.1</sub> composite under 5 T magnetic fields have been calculated as 1.18 10<sup>-3</sup> J kg<sup>-1</sup>K<sup>-1</sup> and 17.8 J/ kg respectively. Therefore, we indicate that the obtained results will represent attention of research for the development of magnetic refrigeration devices

## Method

First of all, two manganites powder with compositions of Pr0.67Ba0.22Sr0.11Mn0.95Fe0.05O3 / Pr<sub>0.67</sub>Ba<sub>0.22</sub>Sr<sub>0.11</sub>Mn<sub>0.9</sub>Fe<sub>0.1</sub>O<sub>3</sub> were elaborated using the solid-solid method at high temperature. Synthesis method is reported elsewhere [19] using stoichiometric amounts with higher purity than 99.9% of Pr<sub>6</sub>O<sub>11</sub>, BaO<sub>2</sub>, SrO<sub>2</sub>, MnO<sub>2</sub> and Fe<sub>2</sub>O<sub>3</sub>. Then, the appropriate amount of both samples with 50% of each was mixed and a homogenous powder was pressed into pellets and sintered at 1200 °C for 12h. The obtained samples were characterized by powder X-ray diffraction at room temperature in Panalytical X'PERT Pro diffractometer, using  $\theta/2\theta$ Bragg-Brentano geometry with diffracted beam monochromatized  $CuK_{\alpha}$  radiation. The diffraction patterns were collected by steps of 0.017 over the angle range 10-70°. Structural analyses were checked by the Rietveld method using the FullProf program [20, 21]. Magnetization measurements were carried out using a vibrating sample magnetometer (VSM) operating between 40 and 240K with an applied magnetic field up to 5T. Magnetocaloric effect (MCE) was estimated based on the magnetization measurements versus magnetic applied field up to 5T at several temperatures.

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### Authors' contributions

K.S: Experimental parts, structural analyses and contribute to writing the manuscript. M.E: magnetization measurement and contribute to writing part of the manuscript

### **Data Availability Statement**

The data that support the findings of this study are available from the corresponding author upon reasonable request.



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